

聚偏二氟乙烯及其共聚物薄膜非线性光学研究进展

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聚偏二氟乙烯及其共聚物薄膜非线性光学研究进展

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摘要:聚偏二氟乙烯 (PVDF) 及其共聚物薄膜拥有极佳的电活性、较高的衍射效率、显著的非线性光学效应, 广泛应用于光电转换、光调控、光开关等光电功能器件等。本文简要介绍了近年来 PVDF 及其共聚物薄膜非线性光学研究方面的主要进展, 指出该类薄膜共混、纳米掺杂、超薄化的发展方向。同时指出需从第一性原理-光子带隙计算着手研究其非线性光学性质, 以高灵敏度 Z-扫描及马克条纹法结合椭偏为主要测量方式。本综述将为该类薄膜的非线性光学研究及制备提供一定的参考。

关键词:聚偏二氟乙烯及其共聚物; 薄膜; 铁电; 非线性光学

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Research progress on nonlinear optics of polyvinylidene fluorid and its copolymers films

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Abstract: Polyvinylidene fluoride (PVDF) and its copolymers films have been extensively used in photoelectric functional devices such as photoelectric conversion, optical regulation, optical switch. They are the most important polymeric ferroelectricity materials with excellent electro-active properties, high diffraction efficiency and significant nonlinear optical effect. We summarize the progress in nonlinear optical effect of polyvinylidene fluoride and its copolymers films both in domestic and foreign research within the last several years. We illustrate that the development direction of the films will be nanoscale-doping, blending modification and ultrathin. The nonlinear optical properties should be investigated by the first-principle and photonic band gap calculations, and measured by the means of the high sensitivity Z-scan, Marker fringe combing with ellipsometry. This study can provide an insight for the development and utilization for polyvinylidene fluoride and its copolymers films in future.

Key words: polyvinylidene fluoride (PVDF) and its copolymers; ultrathin films; ferroelectricity; nonlinear optics (NLO)

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1 引言

铁电体是一类具有自发极化特性的材料,由于其具有压电性 (Piezoelectricity)、热释电性 (Pyroelectric)、铁电性 (Ferroelectricity) 以及电光效应 (Electro Optic Effect, EO)、声光效应 (Acousto Optic Effect)、光折变效应 (Photorefractive Effect) 和非线性光学效应 (Nonlinear Optical Effect, NLO) 等重要特性,吸引了众多科研人员投入该领域研究。目前为止,已制备出超过 200 种铁电体材料。其中,聚偏二氟乙烯 (Polyvinylidene fluoride, PVDF) 及其共聚物因柔韧性好、成本低、易加工,兼具良好的铁电、光学和力学性能,一直是研究人员关注的焦点^[1]。

PVDF 及其共聚物铁电薄膜材料的晶胞结构具有天然非对称性,这导致自发偶极矩的产生,因此拥有较大的压电系数、热释电系数,在居里点附近还存在明显的结构相变。同时, PVDF 及其共聚物晶体的铁电相 (β 相) 的二阶非线性系数接近于石英、铌酸锂,表现出显著的 NLO 效应^[2]。它们还具有较高的衍射效率与光折变效应 (双折射, Δn), 以及宽波长光学窗口 (200~1 200 nm)^[3-4], 上述优点使得以其为基体的高介电常数功能材料体系可制成低成本的微机电系统 (MEMS) 器件,从而广泛应用于光电功能转换、光调控、光探测、光限制等领域^[5-9]。本文将主要介绍近年来 PVDF 及其共聚物薄膜 NLO 方面的研究进展。

2 聚合物非线性光学材料概述

2.1 聚合物非线性光学简介

光束通过透明介质时,介质的原子(或分子)在光场作用下会发生极化。在弱光场作用下产生线性极化,而在强光场作用下,除会产生线性极化外,还会产生二次、三次等非线性极化,表现出 NLO 效应,其电极化强度矢量 P 与光波电场 E 之间的函数关系^[10] 为:

$$P_i(\omega_r) = \varepsilon_0 \chi^{(1)} \times E_m(\omega_j) + \varepsilon_0 \chi^{(2)} \times E_m(\omega_r) E_n(\omega_s) + \varepsilon_0 \chi^{(3)} \times E_m(\omega_r) E_n(\omega_s) E_p(\omega_t) + \dots, \quad (1)$$

其中, $\chi^{(1)}$ 导致折射、反射等线性光学现象,二阶非

线性极化张量 $\chi^{(2)}$ 产生二次谐波 (SHG)、和频、差频、光学整流、线性电光效应、光参量振荡等非线性现象;与 $\chi^{(3)}$ 相关的典型效应为三倍频 (三次谐波, THG), 和、差混频,双光子吸收和简并四波混频。线性电光效应 (EO, $\Delta n = rE$) 可看作特殊的一阶 NLO 效应,而二次电光效应 ($\Delta n = q\lambda E^2$) 则直观表明了折射率与外加电场的二次非线性关系。从张量的性质可知,具有对称中心晶体结构的材料,其二阶 NLO 效应不存在,而任意材料均可能存在三阶 NLO 效应^[11]。聚合物 NLO 材料具有 NLO 效应强、响应速度快 (10^{-14} s 左右)、光损耗阈值高、介电常数小等优点,且材料种类多、结构多样、制备工艺简单,可满足多种要求,因此获得了极大发展。依据非线性光学原理及设计原则,聚合物 NLO 材料可分为非共轭、部分共轭与共轭聚合物 3 大类;从化学组成上看主要有聚丙烯酸酯类、聚苯乙烯类、聚氨酯类、环氧树脂类、聚酰亚胺类等;从结构特点又可大致分为客体-主体聚合物、功能化生色团聚合物 (含侧链与主链)、交联生色团-大分子基体、枝状大分子与自组装超晶格生色团聚合物 5 大类。从诸多文献报道来看,部分 NLO 极化聚合物的单项性能指标已超过 LiNbO_3 ^[12-19]。

2.2 聚合物非线性光学特性测量

简便快捷、精确先进的非线性光学测量技术是聚合物 NLO 材料研究制备及表征的基础。根据聚合物 NLO 材料特性,聚合物 NLO 测量方法可分为二阶非线性测量技术与三阶非线性测量技术两类。前者主要测量二阶非线性材料 (非中心对称材料) 的相关参数,有马克条纹法 (测量微米级厚度薄膜材料) 与电场诱导二次谐波法 (测量纳米级尺度超薄膜);由于理论上任意材料均有三阶非线性响应,后者则包含所有非线性测量技术,其中马克条纹法最为简单常用。按照测量条件或信号产生方式,聚合物 NLO 测量又可分为泵浦探测法、干涉法、简并三波与四波混频及 Z-扫描法^[20-25]。

一般来说, NLO 测量过程包括非线性信号产生与非线性信号探测处理两大部分,须有多光路结构;而 Z-扫描法中 NLO 信号产生与探测由单光束完成,其测量过程得到极大简化。Z-扫描原理如图 1(a) 所示。高斯光照射 NLO 材料时,介

质折射率会发生显著变化;当样品沿 Z 轴相对于焦点 (F') 移动时,由于样品的非线性作用,经小孔后的透射光光强将发生变化。当样品为正 NLO 材料 (自聚焦) 时,由 $-Z$ 向焦点移动时,光强逐渐增强,导致样品 NLO 效应变大,且愈靠近光轴, NLO 效应愈强,反之亦然。文献 [21] 报道的马克条纹技术,可以较为方便地获得聚合物 NLO 材料的二次谐波 (Second Harmonic Generation, SHG) 与三次谐波 (Third Harmonic Generation, THG) 信号,从而计算得出 $\chi^{(2)}$ 、 $\chi^{(3)}$ 等 NLO 参

数值 (图 1(b))。文献 [24] 介绍了一种用于测量弱光 NLO 效应的新方法-质心扫描技术 (图 1(c)),其机理与 Z 扫描相同,特殊之处在于:通过在位置灵敏度探测器前方放置一个不透光的挡板来记录远场光斑横向尺寸变化,从而测量材料的 $\chi^{(3)}$,其灵敏度可达万分之一波长 ($\lambda/10000$)。此外,文献 [25] 使用焦距可调透镜来改进的 F - Z 扫描技术,得到了与传统 Z -扫描等效的非线性折射和双光子吸收现象的实例数据,最大优点在于无需样品位移,测量速度显著提高。

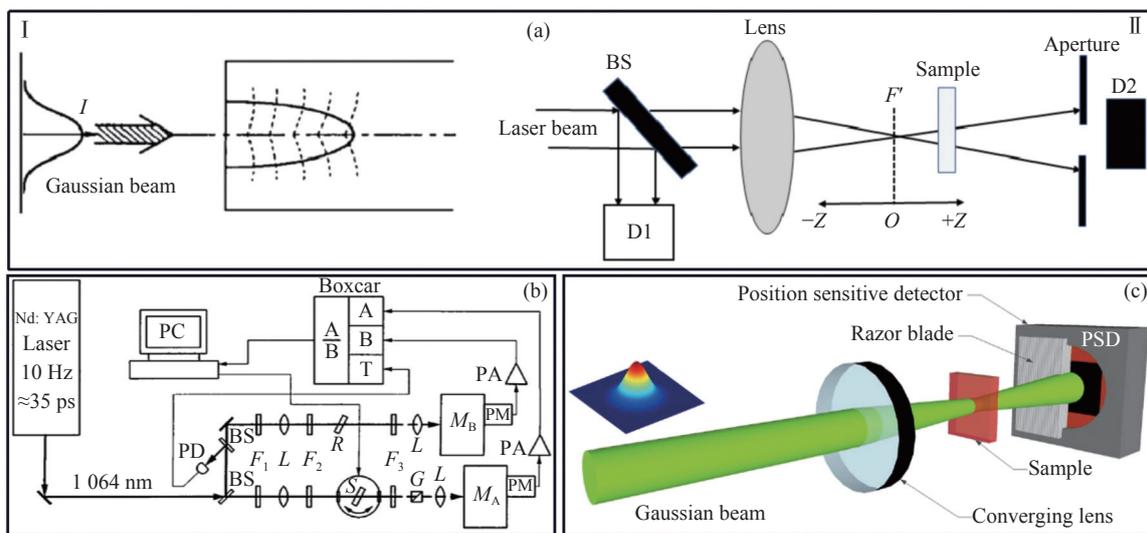


图 1 Z -扫描技术示意图。(a) Z -扫描技术原理: I 表示高斯光进入非线性材料后的波前畸变, II 为 Z -扫描实验装置,其中 D1 与 D2 为光电探测器, D2/D1 比值与 Z 存在函数关系^[20]; (b)通过马克条纹技术测量二阶及三阶非线性参数示意^[21]; (c)质心扫描技术原理^[24]

Fig. 1 Schematic illustration of the Z -scan technique. (a) The principle of the Z -scan technique: I -the wavefront deformation of Gaussian beam when entering nonlinear materials, II -the Z -scan experimental apparatus in which the ratio D2/D1 is recorded as a function of the sample position Z , D1 and D2 are photodetectors^[20]; (b) setup for second and third harmonic generation by means of the Maker fringe technique^[21]; (c) the principle of barycentric scan technique^[24]

总之, Z -扫描技术与马克条纹法为聚合物 NLO 研究提供了快捷简便的测量手段,对其进行持续深入的改进与研究会进一步提升该类材料性能表征的科学性与准确性。

3 PVDF 及其共聚物薄膜的非线性光学特性研究

3.1 PVDF 及其共聚物薄膜概述

依据聚合物非线性光学概念及设计原则,结合目前研究应用现状^[12-19],本综述特定义: PVDF 及其共聚物光学薄膜是指以 PVDF 及其共

聚物为功能基体,利用共聚、共混、掺杂等工艺,采取溶胶凝胶、溶液浇铸、静电纺丝、Langmuir-Blodgett(LB)等方法制备,以光学或光电功能为主的薄膜,其厚度一般在纳米与微米尺度 (1 nm~300 μm)。按 NLO 聚合物材料的化学组成,将 PVDF 及其共聚物光学薄膜分为两类:即本征型 PVDF 聚合物薄膜与掺杂型 PVDF 聚合物薄膜。前者是指以 PVDF 均聚物、PVDF 共聚物及 PVDF 共混物为功能实现主体,以交联、共混等工艺方法制备的薄膜,包含有 PVDF、P(VDF-TrFE)、P(VDF-TeFE)、P(VDF-TrFE-CFE)、PVDF-HFP、PVDF/PMMA、PVDF/PS 等。后者是指将功能化

光电功能填料纳米颗粒或纤维以物理掺杂的方式添加到 PVDF 及其共聚物中制备而成的薄膜, 按照填料性质可分为金属氧化物掺杂、无机非金属晶体掺杂、低维碳材料掺杂、金属盐掺杂及多种填料复合掺杂等几大类 (详见图 2)。特别指出,

由于 PVDF 及其共聚物在线性弱场条件下的光学性质弱于常见的 PMMA、PC 等光学材料, 且线性光学^[11] 可归为非线性光学中的特殊情况。因此, 本综述中 PVDF 及其共聚物光学薄膜的非线性光学研究将涵盖其线性光学 (弱光电场) 范围。

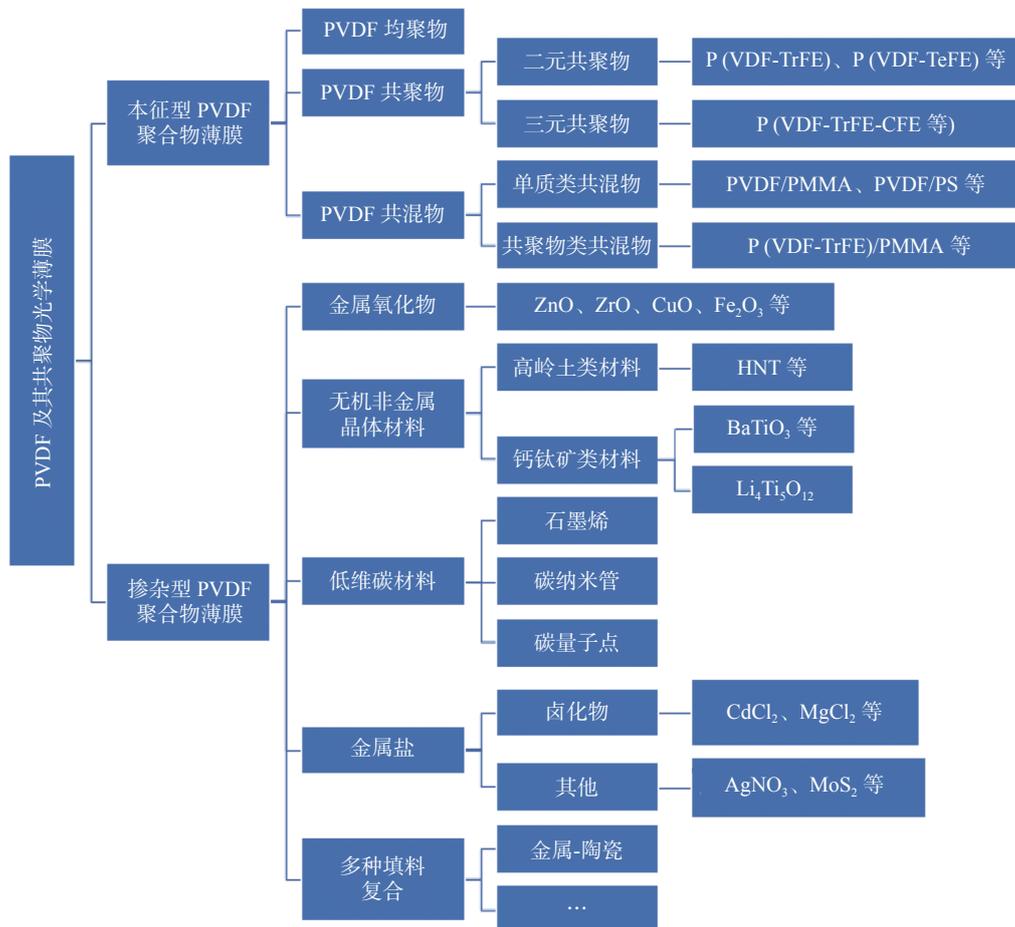


图 2 PVDF 及其共聚物光学薄膜的分类

Fig. 2 Classification of PVDF and its copolymers optical films

3.2 本征型 PVDF 及其共聚物薄膜

综合相关文献可得出, PVDF 及其共聚物光学薄膜研究制备主要有两种技术路线。第一种为: 将偏二氟乙烯 (VDF) 与三氟乙烯 (TrFE) 或四氟乙烯 (TFE) 按照比例共聚, 生成更易自发极化的共聚物大分子, 从而结晶成铁电相, 其 NLO 强度或相关参数与晶相的铁电极化和聚合物的结晶度成正比^[26]。第二种为: 将光学性能较好, 且与 PVDF 类聚合物相容性较好、极性相近的一种或多种聚合物与之按照一定比例进行共混, 也可制备出高 β 相的共混物薄膜, 如 PVDF/PMMA 共混物薄膜^[27]。

早期科学家主要从制备 PVDF 的均聚物薄膜着手, 研究其铁电及 NLO 效应。由于常温下 PVDF(pure) 只是作为半晶聚合物存在, 其组成以 α 相为主, 为获得更具使用价值的极性相- β 相, 则必须通过加温 (退火或淬火)、拉伸、强电场等方法获得^[28-45]。1969 年, Kawai 将 PVDF 薄膜加热至 $100\text{ }^{\circ}\text{C} \sim 150\text{ }^{\circ}\text{C}$ 时, 在淬火+单轴拉伸+高温强电场 (-300 kV/cm) 条件下进行极化, 极化后的 PVDF 薄膜具有强压电效应, 这一发现为研究其 NLO 效应奠定了基础, 并提供了思路与方向^[30]。于是, 从上世纪 70 年代开始, 科学家开展了大量研究, 先后发现了 PVDF 薄膜热释电性质^[46]、光

谱区透射特征^[46]、双折射性质^[46]、SHG 振荡^[4]、电光迟滞回线^[47]、LB 膜的 SHG 效应^[48-49] 等现象,并测试得出 PVDF 均聚物及其共聚物、共混物薄膜的红外区特征谱^[33-34] 及电光系数^[50]、弹光

系数^[50]、二次电光系数^[50]、 n 、 k 等 NLO 及 LO 参数^[51-70],取得了一系列重要成果,梳理总结 PVDF 及其共聚物薄膜的 NLO 性能参数如表 1 所示。

表 1 PVDF 及其共聚物薄膜的光学性能相关参数
Tab. 1 Optical characteristics of PVDF and its copolymer films

Refractive index	second-harmonic coefficients	Intrinsic Birefringences ^[51]			Coherent wavelength	
		^a Fibers	α (II)	β (I)		γ (III)
^a 1.42 ^[51]	d_{33} : 0.22 pm/V ^[52]	0.030 2	^d 0.023 6	0.030 773	0.001 022	
^a 1.425 ^[46]	d_{31} : 0.05 pm/V ^[52]		^e 0.023 6	0.030 773	0.001 022	30 μm ^[46]
^a 1.41-1.49 ^[54]	^b -35 pm/V ^[55]	0.038 7	^f 0.095 0	0.113 2	0.073 9	
^b 1.408 ^[71]	^b d_{33} : -40 pm/V ^[60] ;		37 μm ($\pm 3 \mu\text{m}$) ^[52]			
^b 1.404-1.540 ^[68]	^b d_{33} : -20 \pm 2 pm/V ^[60] ;					
^a 1.99 ^[69]	^c d_{33} : 1.66 pm/V ^[70]					
^a 1.4-1.9 ^[55]	^c d_{31} : 2.01 pm/V ^[70]			^c 0.008 2 ^[70]		
electro-optic coefficients(EO)		Elasto-optic coefficients		quadratic electro-optic coefficients		
$ r_{51}^x $	^[50] 0.10 $\times 10^{-12}$ mV ⁻¹	$ \pi_{11}^E - (n_2^3/n_1^3)\pi_{12}^E $	^[50] 3.6 $\times 10^{-12}$ m ² N ⁻¹	$ g_{44}^x $	0.02 m ⁴ C ⁻² ^[50]	
	^[70] 0.23 $\times 10^{-12}$ mV ⁻¹				21 m ⁴ C ⁻² ^[59]	
$ r_{42}^x $	^[50] 0.21 $\times 10^{-12}$ mV ⁻¹	$ \pi_{21}^E - (n_2^3/n_1^3)\pi_{22}^E $	^[50] 1.8 $\times 10^{-12}$ m ² N ⁻¹	$ h_{55}^x $	6.8 $\times 10^{-23}$ m ² V ⁻² ^[50]	
	^[70] 1.78 $\times 10^{-12}$ mV ⁻¹				4.11 $\times 10^{-18}$ m ² V ⁻² ^[59]	
$ r_{13}^x - (n_2^3/n_1^3)r_{23}^x $	^[50] 0.38 $\times 10^{-12}$ mV ⁻¹	$ \pi_{13}^E - (n_2^3/n_1^3)\pi_{23}^E $	^[50] 1.8 $\times 10^{-12}$ m ² N ⁻¹	$ g_{13}^x - (n_2^3/n_1^3)g_{23}^x $	^[50] 0.01 m ⁴ C ⁻²	
	^[70] 1.48 $\times 10^{-12}$ mV ⁻¹				$ h_{13}^x - (n_2^3/n_1^3)h_{23}^x $	^[59] 26 $\times 10^{-23}$ m ² V ⁻²

a: Preparation with Spin-coating; b: Preparation with LB method; c: Calculations based on the formalism of Hughes and Sipe; d: Calculations based on Bunn, Denbigh (C-C, C-H) and Denbigh (C-F); e: Calculations based on Bunn, Denbigh (C-C, C-H) and Vogel(C-F); f: Calculations based on Denbigh (C-C, C-H) and LeFevre and LeFevre (C-F). g: Birefringence of Melt-Spun PVDF Fibers before (above) and after (below) the poled films (Clod-Drawing, Elongated-40%)

一些经典研究时至今日仍值得持续关注学习。如文献 [46] 中用巧妙的楔形膜实验得出 PVDF 薄膜的热释电性质与双折射以及膜厚存在显著相关^[4](图 3(a)),并表现出 SHG 振荡 (图 3(d)),同时 PVDF 的 NLO 系数可参照 SiO₂ 算出。更为难能可贵的是,通过一系列实验计算得出了 PVDF 的 EO 系数 (r_{ijl}^x)、二次 EO 系数 (g_{ijk}) 及介电系数值,并与氧八面体铁电体 (如钛酸钡) 等进行了定量比较^[46]。1989 年, A. Wicker^[52] 等首次实验测试出了 P(VDF-TrFE) 的折射率及 SHG 系数、色散 (双折射) 等重要光学参量,最为重要的是,该文献中将丁香酚 ($n=1.525$) 作为折射率匹配液,入射角调整到近乎平行于薄膜 (67.8°),观察到了 NLO 效应中的又一重要现象-相位匹配 (图 3(c))。而 Moti Ben-David^[54] 采用椭圆法详细研究了 P(VDF-TrFE-CTFE) 共聚物与 P(VDF-TrFE-CTFE)/PMMA 共混物薄膜的光学性质,发现:在 200~1000 nm 宽波长范围内,微米级厚度 (200~2500 nm) 的 P(VDF-

TrFE-CTFE) 薄膜近乎透明 (n 为 1.49~1.4, $k < 0.016$),其 n 、 k 值与经典的 Cauchy 与 Sellmeier 模型相吻合。

现有文献报道来看,厚度为 μm 级的二维 PVDF 薄膜材料研究较为充分^[60-64],而 nm 级的 PVDF 超薄膜或单分子膜可能会有更为显著的铁电性与 NLO 效应^[65-66]。LB 薄膜是将不溶解的成膜材料分子在水-气界面上紧密有序排列后,形成单分子膜,然后再转移 (沉积) 至固体衬底上叠加形成的单分子膜^[67]。1995 年,俄罗斯科学家 Palto^[68] 等成功制备出了 P(VDF-TrFE) (70:30 mol%) 的 LB 超薄膜 (15nm、30 层),并观察到了薄膜的热释电性能。由于 PVDF 的 LB 膜单分子层极薄 (nm 级)、范德华力键合较弱,在 LB 膜层表面沉积电极往往会造成表面结构破坏或损伤,导致测试失真。Palto 团队^[48-49] 建立了 P(VDF-TrFE) 的 LB 铁电膜结构模型,充分利用 PVDF 本征极化特性导致的 NLO 典型效应-SHG,作为“探针”来形象表征其晶体结构及相变特性。此外,近年来有

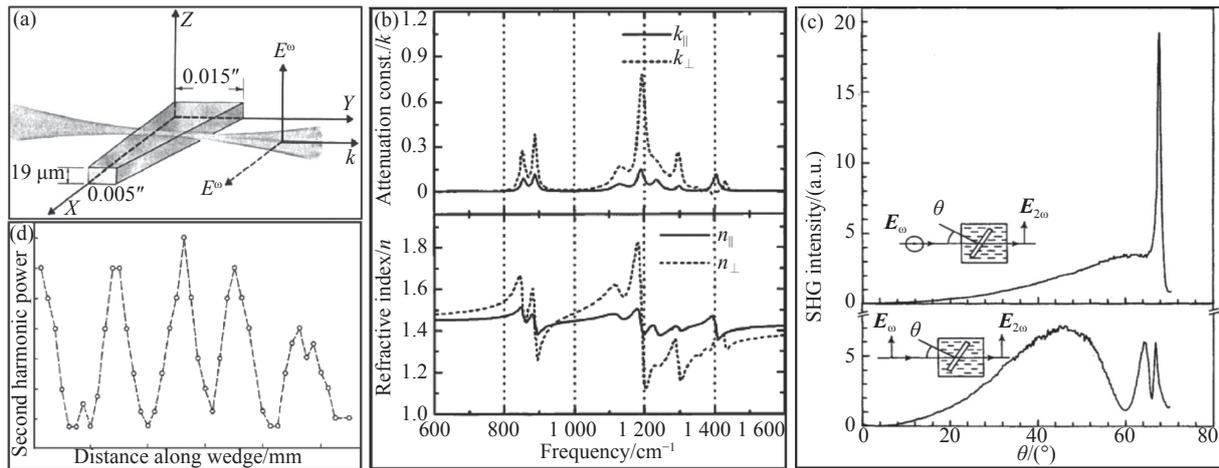


图3 PVDF及其共聚物光学薄膜非线性光学经典研究示意图。(a)用于二次谐波测试的PVDF楔形薄膜的几何形状, k 表示入射激光传输方向^[4];(b)PVDF的LB膜红外-椭偏谱中测得的消光系数 $k_{||}$ 、 k_{\perp} 与折射率 $n_{||}$ 、 n_{\perp} ^[57];(c)通过绕垂直轴旋转平面样品获得的马克条纹,入射光为水平偏振光。上图:当入射偏振光垂直于丁香酚溶液,在67.8°入射角下的相位匹配证据,二次谐波仍为水平方向;下图:在注满丁香酚溶液的透明试管中,通过绕垂直轴旋转极化共聚物膜获得的马克条纹,入射偏振光和二次谐波均为水平方向^[52];(d)当楔形样品在平行于激光束方向上平移时获得的SHG振荡条纹图^[4]

Fig. 3 Schematic illustration of classical research on nonlinear optics of PVDF and its copolymer optical thin films. (a) Sample geometry and orientation for SHG measurements on PVDF film. The direction of the laser beam propagation corresponds to k ^[4]. (b) The attenuation constants $k_{||}$ and k_{\perp} and the refractive indices $n_{||}$ and n_{\perp} of the LB films obtained from the IR-VASE (Variable angle spectrometer ellipsometry) data analysis^[57]. (c) Maker fringes obtained by rotating the planar samples around a vertical axis, with a horizontal polarization of the incident beam. Above: evidence of phase matching at an incidence angle of 67.8° in Eugenol when the incident polarization is vertical. The SHG polarization is still horizontal. Below: Maker fringes obtained by rotating the poled copolymer film around a vertical axis in a transparent cell filled with Eugenol. The incident polarization and the SHG polarization are horizontal^[52]. (d) SHG fringes pattern obtained when wedge sample is translated parallel to its length across laser beam (fundamental)^[4]

不少研究人员尝试采用无损、常规的光学测试方法来研究纳米尺度的PVDF共聚物薄膜性质。如 Bai M^[57-58]与 Wang J L^[71]等利用椭偏技术研究发现:PVDF及其共聚物、均聚物的LB膜在可见光、近红外波段均透明,聚合物分子链在薄膜平面内为自由取向,在平行及垂直于薄膜表面两个方向却表现出色散、弱吸收等薄膜典型特征(图3(b)),其 n 、 k 等测试数据可以采用Cauchy、Sellmeier及Lorentz等模型拟合。这些研究均表明PVDF及其共聚物薄膜在微米、纳米尺度均表现出类PC(聚碳酸酯)、PMMA(聚甲基丙烯酸甲酯)的色散、吸收等线性光学性质,而特殊之处在于其本征极化性能导致的NLO效应,LB方法则为这一特性的表征提供了实现手段。

3.3 掺杂型PVDF及其共聚物薄膜

一般来说,在介观及微观尺度下,材料不均匀性是显而易见的。引入两种或多种材料,在一定

温度、压力条件下进行掺杂来获得性能独特的复合材料已成为材料设计制备的重要方式。大多数无机铁电材料力学性能差、声阻抗高、击穿强度低,但有着较高的介电常数;而聚合物铁电材料(PVDF)具有机械性能良好、易于合成、低声波阻抗等优点,但缺点是与无机晶体材料相比,介电常数不高。将两者结合,将NLO性能更优的氧化物晶体或非晶体无机材料等纳米化后掺杂在PVDF及其共聚物或共混物中,通过溶胶凝胶等常规方法制备生成纳米聚合物薄膜(Polymer Nano Composites, PNC)是提高或改善其非线性光学性能的现实路径之一。目前,掺杂PVDF及其共聚物光学薄膜主要有金属氧化物材料、无机非金属晶体材料、碳材料、金属盐、其他材料等几大类(详见图2)。

3.3.1 金属氧化物

ZnO、CuO、ZrO₂、TiO₂等金属氧化物,作为

电学与光学性能俱佳的半导体材料, 由于具有较低或较宽的光学带隙, 且光学性能突出, 一直被公认为是优良的光学晶体掺杂材料, 将其纳米化后添加至 PVDF 及其共聚物薄膜中, 理论上可以降低薄膜的活化能, 改善聚合物的线性及非线性光学性能。其中, ZnO 纳米颗粒由于其极高的化学活性及优异的光催化活性, 宽禁带、高折射率和紫外吸收等独特性质而备受研究人员关注^[72]。PVDF/ZnO 薄膜, 随着 ZnO 含量的增大, 薄膜透过率降低、吸收增强, 带隙显著降低 (图 4(c)), 而这些是由于 ZnO 纳米颗粒导致的 PVDF-ZnO 纳米复合材料的缺陷增多、局域态密度增大, 进而使导带中载流子增多而降低了能隙, 并伴随有瑞利散射的增强而使薄膜更透明 (可见光波段)。而单光束 Z-扫描测试表明其具有典型的负

NLO 效应-自散焦效应与负非线性折射率特征 (图 4(d))^[73-75]。

尽管 ZnO 纳米颗粒的光学性能优异, 但其较高的电阻率影响了其综合性能的发挥; 同时, 对于基体聚合物来说, PVDF 均聚物熔态下掺杂物的分散性较差, 熔体粘度高, 其半结晶性质导致透明度低。因此, 研究人员尝试从聚合物基体及纳米颗粒改性两个方向, 深度探究聚合物/纳米 ZnO 薄膜的光学性能及制备。对 ZnO 颗粒进行功能化改性, 如掺杂金属元素^[76]使其晶体结构发生位置失配以降低其带隙; 将 PVDF 与透明性更好、力学性能优异的聚合物共混, 如 PVDF/PVC^[77]、PVDF/PMMA^[76,78-81] 等来优化基体聚合物综合性能 (图 4(a))。

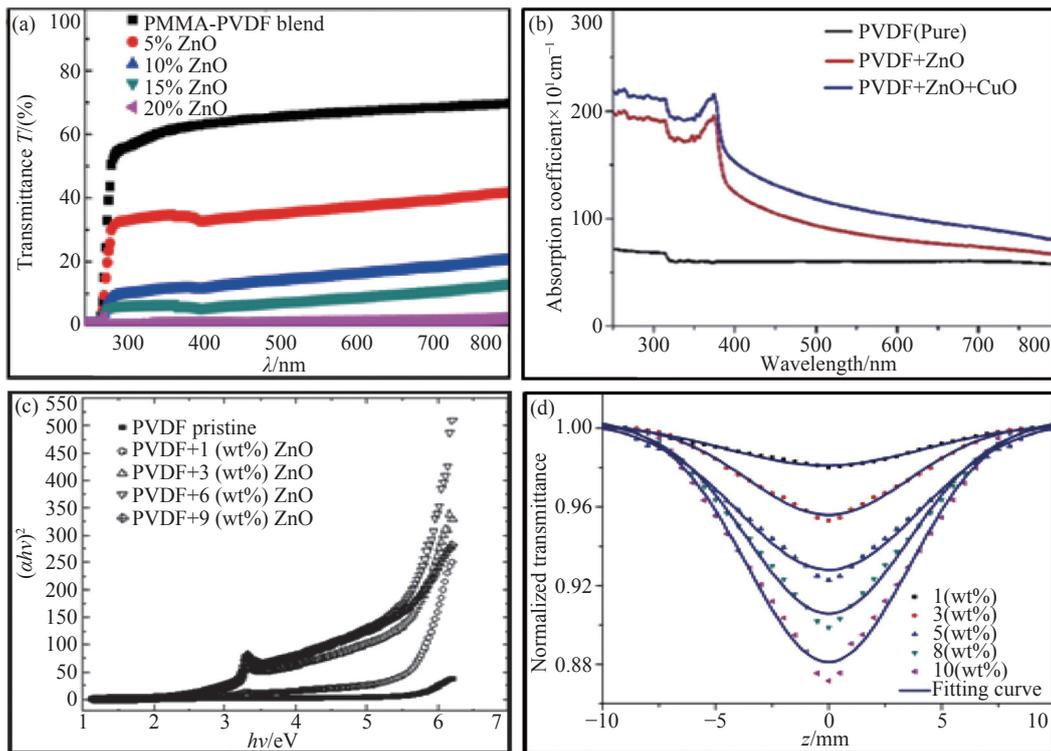


图 4 PVDF/金属氧化物纳米复合薄膜的非线性光学性能。(a) PMMA/PVDF-ZnO 纳米聚合物薄膜的透射率变化^[78]; (b) PVDF/ZnO/CuO 纳米聚合物薄膜线性吸收系数光谱^[81]; (c) 单一 PVDF 及 PVDF-ZnO 纳米薄膜的直接带隙^[73]; (d) PVDF/ZnO 纳米复合薄膜在开孔 Z-扫描模式下的归一化透射率与样品位置的关系^[74]

Fig. 4 NLO properties of PVDF/metallic oxide nanocomposites films. (a) The transmittance of PMMA/PVDF-ZnO nanocomposites^[78]; (b) linear absorption coefficient spectra of PVDF/ZnO/CuO nanocomposites^[81]; (c) plots (direct band gap) for PVDF pristine and PVDF-ZnO Nanocomposites^[73]; (d) the normalized transmittance as a function of sample position in open-aperture Z-scan for PVDF/ZnO nanocomposites^[74]

单质 ZnO 纳米颗粒的添加使纳米聚合物薄膜的非线性折射率得到提升, 而多种金属氧化物

类铁电晶体的掺杂, 可能会使薄膜的 NLO 性能得到更大改善。CuO 晶体为典型的窄能隙 *p*-型

半导体, 其室温下的能隙宽度约为 (1.2~1.4) eV, 与 ZnO 相近, 其 NLO 性能同样出色^[79-80]。H. M. Shanshool 团队^[81]发现: CuO 纳米颗粒的加入, PVDF/ZnO/CuO 薄膜的线性吸收系数增大 (较 PVDF/ZnO)、透射率降低、带隙减小。开孔状态下 Z-扫描测试结果显示: 其非线性吸收更加强烈 (图 4(b), 彩图见期刊电子版), 表明 CuO 的加入, 使薄膜能隙降低, 同时非线性吸收率、非线性折射率 (绝对值) 及三阶非线性极化率均增大。

此外, ZrO₂^[69]、Al₂O₃^[82]、SnO₂^[82]、TiO₂^[82]、

Cr₂O₃^[83]、HfO₂^[84]、La₂O₃^[85]、Fe₂O₃^[86]、Fe₃O₄^[87]、Nd₂O₃^[88]、Gd₂O₃^[89] 等金属氧化物纳米颗粒的一种或多种复合掺杂到 PVDF 及其共聚物或共混物薄膜中, 也能使其光学吸收性能出现变化。这些金属氧化物纳米颗粒固有的物理形态结构与介电特性, 使得掺杂后纳米聚合物薄膜的光学、电学等性能参数发生变化, 导致掺杂后薄膜吸收率增大 (或降低)、光学带隙降低 (或增大), 并产生光电导、光热、荧光吸收等效应, 其掺杂后的光学性能参数见表 2。

表 2 PVDF-金属氧化物纳米聚合物薄膜的线性及非线性光学相关参数

Tab. 2 Linear optical and nonlinear optical parameters of PVDF/MO (metallic oxide) nanocomposites films

Nonlinear optical parameters									
Samples	L_{eff}	β	ϵ_r	n_2	$\Delta\Phi_0$	E_d (eV)	E_i (eV)	E_a (eV)	$ \chi^{(3)} $
P/ZnO ^[81]	1.051 ^[81] 0.2339~0.30405 ^[73]	1.942 ^[81] 17.58~2.064 ^[73]	3.15~4.91 ^[81]	-1.624 ^[81] -4.562~ 12.22 ^[73]	0.181 ^[81] 0.147~ 0.29 ^[74]	5.57~4.95 ^[73] 3.24 ^[81]	4.76~3.35 ^[73]	1.16 ^[73]	0.7145 ^[73]
P/ZnO/CuO ^[81]	0.862	4.740	—	-3.220	0.294	—	—	—	1.4039
Samples	β'	ϵ_r	n	p	E_d (eV)	E_i (eV)	A		
P/ZrO ₂ ^[69]	3396.8~26.10	3.97~5.83	1.99~2.41	1.62~7.27	1.53~5.89	2.94~0.76	0.34~9.29		
Linear optical parameters									
Samples	ϵ_r	n	k	Samples	T	ϵ_r			
P/CrO ₂ ^[83]	2.77~5.83	1.395~1.43	0.005~0.03	P/Gd ₂ O ₃ : Eu ³⁺ ^[89]	82%~85%	7~7.5			
				P/PPO/POPOGd ₂ O ₃ :Eu ³⁺ ^[89]	75%~77%	12~15			
Samples ^[82]	α	ϵ_r	E_d (eV)	Samples ^[76]	α	E_d (eV)	E_i (eV)		
P/PEO-Al ₂ O ₃	0.99	3.56~6.35	4.91	P/PMMA-V-ZnO	0.88~0.60 ^[78]	2.8	2.5		
P/PEO-SnO ₂	0.99	2.7~4.68	4.62	P/PMMA-S-ZnO	0.82~0.60 ^[78]	3.0	2.8		
P/PEO-TiO ₂	0.982	3.08~5.05	3.53	P/PMMA-Dy-ZnO	0.78~0.60 ^[78]	2.5	2.2		
Samples	α	n	E_d (eV)	E_i (eV)	$k(10^{-2})$	T	ϵ_r		
P/PMMA-ZnO ^[78]	0.975~0.602 ^[78] 0.83~0.40 ^[76] 0.44 ^[78]	1.29~1.54 ^[78]	5.22~5.75 ^[78] 2.9 ^[76] 5.85 ^[78]	2.6 ^[78]	0.018~0.06 ^[78]	35%~40% ^[78]	2.37~1.67 ^[78]		
P/PMMA ^[78]	0.50~0.35 ^[76]	1.21~1.22 ^[78]	4.4 ^[76]	3.7 ^[76]	0.01 ^[76]	65%~70% ^[76]	1.46 ^[76]		
Samples	α	ϵ_r	E_d (eV)	E_a (eV)	Samples	α			
La ₂ O ₃ /P-TrFE ^[85]	0.968~0.996	163	3.15~2.80	¹ 0.20~0.15; ² 0.84~0.41	P/Nd ₂ O ₃ ^[88]	0.90			
				—	Fe ₃ O ₄ /P-HFP ^[87]	0.90			

Note: P-PVDF, 1- ac activation energy (E_{ac}), 2-dc activation energy (E_{dc}), E_d -Direct band gap, E_i -Indirect optical band gap, n -linear refractive index ($\lambda \approx 633$ nm), E_a -optical activation energy, E_f -Fermi energy, n_2 -nonlinear refractive index ($\text{cm}^2/\text{W} \times 10^{-13}$), β -nonlinear absorption coefficient (two-photon, $\text{cm}/\text{W} \times 10^{-8}$), α -linear absorption coefficient, transmittance, ϵ_r -dielectric constant, $\chi^{(3)}$ -third order nonlinear optical susceptibility (10^{-6} esu), L_{eff} -effective length of the sample (10^{-3} cm), $\Delta\Phi_0$ -the nonlinear phase shift, k -the extinction coefficient ($\lambda \approx 633$ nm), p -average polarizability ($\text{C}^2 \text{ m}^2/\text{J}^{-1} \times 10^{-39}$), β' -first hyperpolarizability ($\text{C}^3 \text{ m}^3/\text{J}^{-2} \times 10^{-51}$), A -anisotropy ($\text{C}^2 \text{ m}^2/\text{J}^{-1} \times 10^{-39}$), V-ZnO: ZnO doped with vanadium, S- ZnO: ZnO doped with sulfur, Dy-ZnO: ZnO doped with dysprosium.

总体来看, 金属氧化物填料的掺杂, 使得 PVDF 及其共聚物薄膜的透射率降低、折射率与消光系数变大, 光子带隙减小, 显著改善或提高了

纳米聚合物薄膜的 NLO 性能。因此, ZnO、CuO 等金属氧化物纳米填料的掺杂, 将成为降低 PVDF 及其共聚物薄膜透射率, 提高其非线性自

散焦效应的重要途径。

3.3.2 低维碳材料

低维碳材料主要指富勒烯 (0 维)、碳纳米管 (一维)、石墨烯 (二维)、碳量子点 (0 维) 等碳元素新型同素异形体材料。因其独特的几何结构和以 sp^2 杂化轨道为主的成键结构, 集优异的电学、力学、热学、光学等性能于一体, 被认为是过去 30 年来材料科学领域最重要的科学发现^[90]。目前来看, 具备 NLO 掺杂价值的低维碳材料主要包含石墨烯、碳纳米管 (多壁或单壁)、碳纳米管量子点 3 类。

石墨烯 (RGO, 还原氧化石墨烯)^[91], 具备特殊的单层二维蜂窝状晶格结构, 带隙极低 (0~0.25 eV), 在较宽波长范围内吸收率约为 2.3% (几乎透明), 同时具有极宽的光谱吸收范围 (可调谐至太赫兹范围), 表现出优异的 NLO 效应^[92]。碳

纳米管 (Carbon Nanotubes, CNTs) 作为一维纳米材料的典型代表, 对远紫外至远红外电磁波均有很强的吸收特性, 光致发光性能良好, 经过分散处理的单层碳纳米管的光致发光效率高达 20%^[93]。碳量子点 (CQDs) 作为 0 维碳纳米材料, 同样具有很强的紫外吸收能力, 光致发光、电致发光效应显著, 而且表现出优异的非线性光学特性-上转换光致发光 (反斯托克斯)^[94]。

目前为止, 大量研究集中于这两种低维碳材料掺杂 PVDF 薄膜后的介电、压电、热释电等性能, 且研究结果均表明掺杂后 PVDF 薄膜的 β 相含量显著增多, 介电常数相应变大^[95-98]。与此同时, 也有学者专注于研究掺杂后的光致发光及紫外吸收等光学效应。掺杂 RGO 的 PVDF 聚合物薄膜其透射率显著降低^[99-100], 而非线性吸收系数则变大, 表现出显著的光限幅效应 (图 5(a)、5(b))。

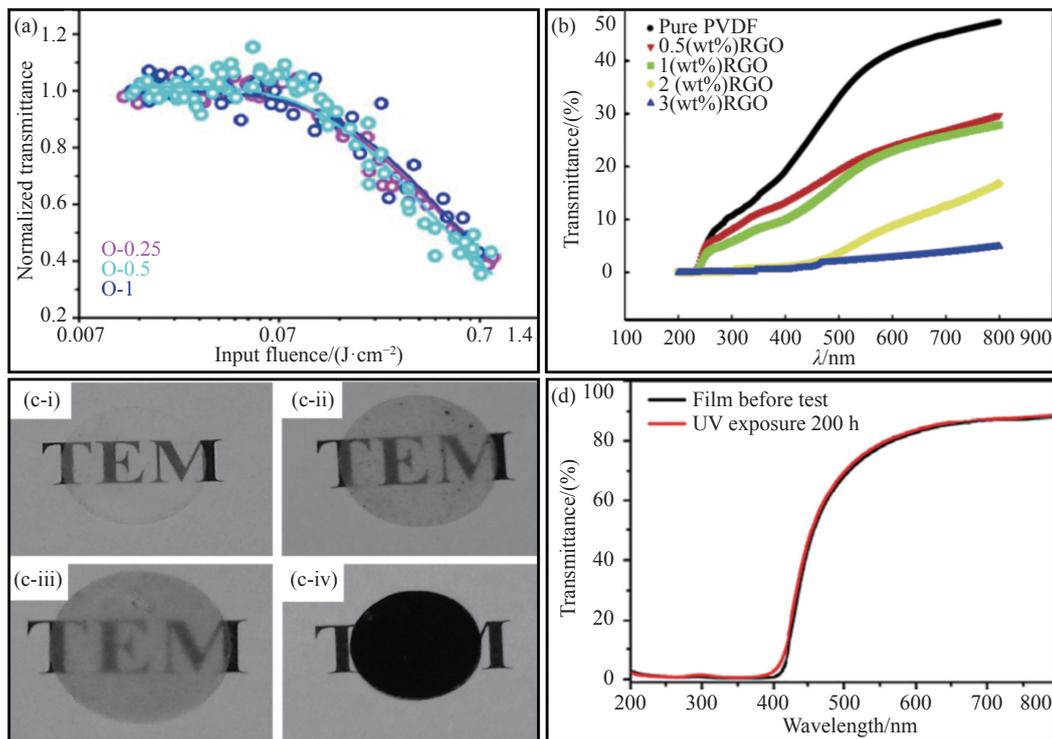


图 5 PVDF/低维碳材料薄膜的非线性光学性能。(a)不同氧化还原石墨烯浓度下的 PVDF/RGO 薄膜光限制图^[93]; (b)单一 PVDF 与 PVDF/RGO 纳米聚合物薄膜的紫外-可见光透射谱^[100]; (c)不同多壁碳纳米管掺杂浓度下的 PVDF/多壁碳纳米管复合薄膜透明照片; (i)单一 PVDF, 质量分数分别为 (ii) 1%, (iii) 2%, (iv) 5%^[101]; (d)CQDs 掺杂的 PVDF 改性薄膜在紫外辐照 200 h 前后透射率对比图^[104]

Fig. 5 NLO properties of PVDF/low-dimensional carbon materials films. (a) Optical limiting graphs for PVDF/RGO films with different concentrations of RGO^[93]; (b) transmittance of pristine PVDF and PVDF-RGO nanocomposites^[100]; (c) transparency camera image for PVDF/MWCNT composites films^[101] with different concentration of MWCNT: (i) pure PVDF, quality score is (ii) 1%, (iii) 2%, (iv) 5%; (d) transmittance of the modified CQDs/PVDF nanocomposite films before and after 200 h of UV exposure^[104]

将不同含量的多壁 CNTs^[101] 添加到 PVDF 可观察到透过率有明显变化: 透明-半透明-模糊-不透(图 5(c)), 透过率显著降低。双壁 CNTs^[102] 的加入使 PVDF-POPD 类纤维结构表面的 PL 效应锐减。Shubnikov 将单壁 CNTs 掺杂在 P(VDF/TrFE) 均聚物中, 也观察到了显著的光折变效应及光伏效应^[103]。利用水热法制备 CQDs 掺杂的 PVDF 薄膜复合材料^[104-105], 其透过率在紫外波段 (200~400 nm) 极低 (图 5(d)), 而在可见光波段迅速增大, 直至接近 PVDF 均聚物薄膜 (400~800 nm);

后者进行 200 h 紫外暴露下透过率基本不变, 表现出很好的抗紫外稳定性 (图 5(d))。

总之, 包括 RGO、CNTs、CQDs 在内的低维碳材料, 其优异的光学及电学性能导致掺杂后薄膜光学带隙与透射率降低、吸收率增大, 表现出优异的 NLO 与紫外屏蔽性能, 同时制备工艺简单、易于实现, 成为研究制备 PVDF 及其共聚物紫外屏蔽薄膜的重要方向。梳理其相关性能参数见表 3 所示。

表 3 PVDF/低维碳材料纳米聚合物薄膜的线性及非线性光学相关参数

Tab. 3 Linear optical and nonlinear optical parameters of PVDF/low-dimensional carbon materials nanocomposites films

Nonlinear optical parameters					
samples	β	n_2	P_{th}	$\text{Im}\chi^{(3)}$	$ \chi^{(3)} $
P/RGO ^[99]	195-400	3.410-6.270	8.4-7.84	6.19-12.95	6.2-12.96
Linear optical parameters					
samples	α	n	E_d (eV)	E_i (eV)	T
P/RGO ^[100]	83%-99%	1.8-2.2	5-4.3	4.4-3.2	30%-1%
PVDF/CQDs ^[105]	90%-98.5%	1.22-1.55	2.96-5.00	1.16-4.32	4%-12%
samples	T				
P-OH@CQDs/PVA ^[104]	88%(300-800 nm)				
PVDF/MWCNT ^[101]	0(5%CNT)、22%(2%CNT)、48%(1%CNT)				

Note: P-PVDF, E_d -direct band gap, E_i -indirect optical band gap, n -linear refractive index ($\lambda \approx 633$ nm), n^2 -nonlinear refractive index ($\text{cm}^2/\text{W} \times 10^{-10}$), β -Nonlinear absorption coefficient (two-photon, cm/GW), α -linear absorption coefficient, T -transmittance ($\lambda \approx 633$ nm), $\chi^{(3)}$: third-order nonlinear optical susceptibility (10^{-11} esu), $\text{Im}\chi^{(3)}$: The imaginary part of third-order nonlinear optical susceptibility (10^{-11} esu), P_{th} -optical limiting threshold power (MW/cm^2).

3.3.3 无机非金属晶体材料、金属盐及多填料复合

种类繁多的无机非金属晶体材料, 为制备性能多样的聚合物薄膜提供了广阔的材料来源。目前为止, 先后有高岭土^[106](黏土之一种)、埃洛石^[107](HNT, 黏土之一种)、硅藻土^[108]、微晶纤维素^[106]、钛酸钡 (BaTiO_3)^[109-110]、尖晶石 ($\text{Li}_4\text{Ti}_5\text{O}_{12}$ 钛酸锂)^[111]、石英 (SiO_2)^[112-113] 等材料被用来与 PVDF 进行掺杂制备光学薄膜器件。其中, 值得关注的是高 β 相含量 (83%) 的 PVDF/HNT 纳米复合薄膜^[107](图 6(a)), 其归一化透射率呈现出先谷-后峰的趋势 (图 6(c)), 添加 HNT 填料后材料导热系数变大 ($0.19 \text{ Wm}^{-1} \text{ K}^{-1}$ 变为 $0.65 \text{ Wm}^{-1} \text{ K}^{-1}$), 其非线性折射率 n_2 及非线性极化率 χ^3 表现出热透镜效应, 同时二者由负转正, 呈增大趋势。

BaTiO_3 作为经典的铁电体材料, 制备生成的 PVDF/ BaTiO_3 薄膜^[109-110], 其退火温度与薄膜厚度均影响了薄膜的带隙及吸收, 温度越高、薄膜越厚, 其带隙越低、吸收越强。同时介电常数显著增大, 表明其极化能力得到提升, 从 8.9(均聚物 PVDF) 变为 26.7 (0.6PVDF/0.4 BaTiO_3 共聚物)。钛酸锂 ($\text{Li}_4\text{Ti}_5\text{O}_{12}$) 为尖晶石结构, 晶格常数与体积变化很小 ($<1\%$), 被称为“零应变”电极材料。文献 [111] 介绍随着 $\text{Li}_4\text{Ti}_5\text{O}_{12}$ 纳米颗粒含量的增多, PVDF/ $\text{Li}_4\text{Ti}_5\text{O}_{12}$ 薄膜光学带隙变小, 同一波长下, 其透过率变小、折射率变大 (图 6(d)), 非线性折射率与三阶非线性极化率均变化很小 (近乎常数), 但是同一波长下随着掺杂浓度的增加, NLO 效应则明显增强。其光学性能参数见表 4 所示。

金属盐多为离子型化合物,其阴阳离子的键合主要是由库仑力-离子键相结合,离子化合物通常熔点和沸点较高,熔融、电离时可导电。近年来,有研究人员先后将氯化铬(CdCl_2)^[114]、氯化镁(MgCl_2)^[115]、 $\text{CuCl}_2/\text{MnCl}_2$ ^[116]、二硫化钼(MoS_2)^[117]、高氯酸锂(LiClO_4)^[118]、硝酸银(AgNO_3)^[119]与PVDF或其共聚物掺杂,得到不少有价值的研究成果。与前述低维碳材料等类似,以金属盐作为填料,均会使薄膜在紫外与可见光区域吸收率与光学带隙降低(图6(e)),但PVDF/ CdCl_2 薄膜 E_g 却反常增加。尤其是文献[118]报道的PVDF-HFP/ LiClO_4 电解质薄膜(掺杂剂重量比为90:10),其采用辐照工艺制备,辐照作用导致薄膜的缺陷增多,引起带隙进一步降低,紫外-可见光谱与荧光谱均出现吸收边红移与展宽。

借鉴复合材料相关理论,也有研究人员尝试将前述金属氧化物、碳纳米管、金属盐及无机非金属材料中一种或几种进行组合设计,将其作为复合填料掺杂到PVDF及其共聚物、共混物基体中制备薄膜,使其在性能上取长补短,产生

协同效应,从而满足各种不同要求。先后有Ag/ BaTiO_3 ^[120]、 $\text{BaTiO}_3\text{-Ag/MWCNTs}$ ^[121]、 KNNLS-CA-ZnO ^[122]、 $\text{TiO}_2\text{@MWCNTs}$ ^[123]、BZT/水泥^[124]、 Ag/ZnO ^[125]等复合填料与PVDF掺杂,由于填料经过处理后其特殊的表面结构或基团(如-OH)与PVDF中H或F原子的强相互作用,使其能在聚合物基体中得到很好分散,其介电常数、 β 相含量均显著增大(图6(b)、6(f))。文献[126]则报道了一种新型复合填料 $\text{CdS/Bi}_2\text{WO}_6/\text{ZnO}$ 掺杂PVDF制备的复合薄膜,三元催化剂填料使薄膜表面粗糙度增加,提供了填料在基体中更多的结合位置,有效地提高了光生电子的迁移效率,其光降解的效率大幅提高。另有文献[127]介绍了将反应型紫外线吸收剂2-羟基-4-(3-甲基丙烯酸酯基-2-羟基丙氧基)二苯甲酮(BPMA)与甲基丙烯酸甲酯(MMA)共聚合成紫外线吸收剂P(MMA-co-BPMA)与PVDF共混得到强紫外吸收复合膜,其在200~345 nm范围内透过率可降至0.4%以下,表现出极好的紫外屏蔽效果。

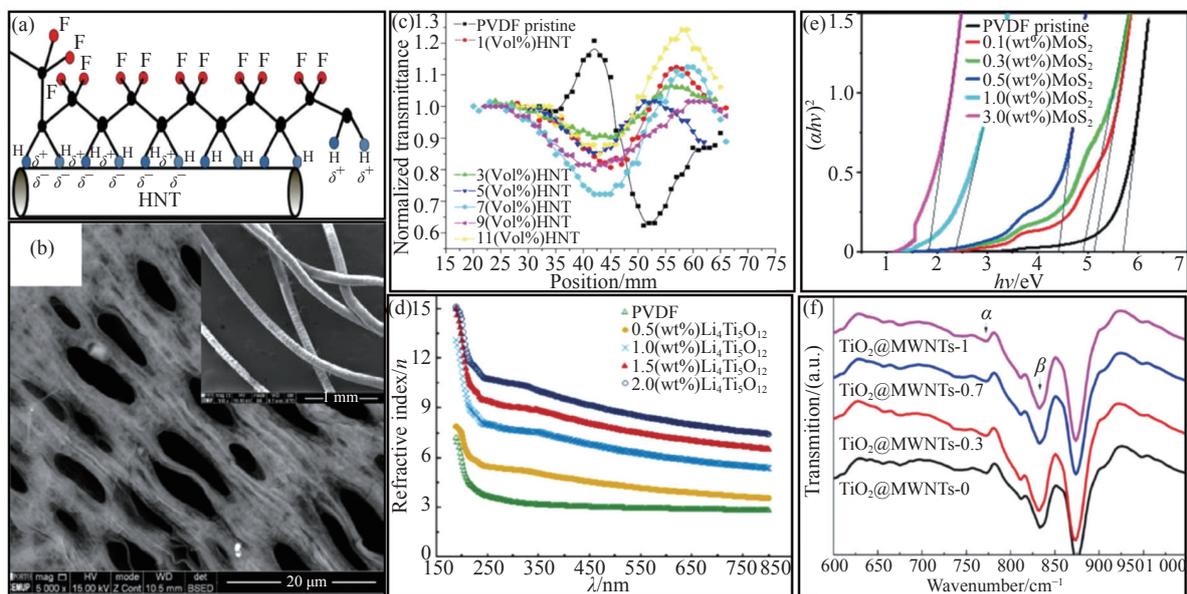


图6 无机非金属材料、金属盐及复合填料掺杂的PVDF薄膜非线性光学特性。(a)PVDF(TTTT)接枝在埃洛石纳米管上的构型示意^[107];(b)PVDF@ SiO_2 @S1扫描电镜图^[113];(c)PVDF/埃洛石纳米管复合薄膜的归一化透射率(不同掺杂浓度下Z-扫描曲线)^[107];(d)单一PVDF与PVDF/钛酸锂纳米复合薄膜的折射率变化图^[111];(e)不同浓度 MoS_2 掺杂的PVDF薄膜直接带隙曲线^[117];(f) TiO_2 掺杂多壁碳纳米管/PVDF复合薄膜的傅里叶红外透射谱^[123]

Fig. 6 NLO properties of PVDF films doped with inorganic nonmetallic crystals, metallic salts, and composite fillers. (a) Schematics of TTTT(PVDF) configuration on HNTs^[107]; (b) SEM micrographs of PVDF@ SiO_2 @S1^[113]; (c) transmittance of the PVDF/HNTs films with Overlaid Z-scan curves^[107]; (d) refractive index for pure PVDF and $\text{Li}_4\text{Ti}_5\text{O}_{12}$ /PVDF nanocomposites^[111]; (e) direct bandgap of the MoS_2 doped in PVDF nanocomposite samples^[117]; (f) FTIR spectrums of TiO_2 @MWCNTs/PVDF composites^[123]

表 4 PVDF/无机非金属晶体材料聚合物薄膜的线性及非线性光学相关参数

Tab. 4 Linear optical and nonlinear optical parameters of PVDF/inorganic nonmetallic crystalline materials nanocomposites films

Nonlinear optical parameters						
samples	$\Delta\phi_0$	L_{eff}	n_2	χ^3		
Pristine PVDF ^[107]	2.05	24.3	-3.13	-1.7872		
P/HNT ^[107]	0.59-2.2	13.9-20.4	1.24-4.89	1.2075-8.9922		
samples	n_{∞}	$\lambda_0(\text{nm})$	S_0	ϵ_{∞}	N/m^*	ω_p
$\text{Li}_4\text{Ti}_5\text{O}_{12}$ ^[111]	2.78	213.62	1.47	8.15	0.472	4.09
P/ $\text{Li}_4\text{Ti}_5\text{O}_{12}$ ^[111]	3.50-7.32	269.68-291.60	1.55-6.19	15.64-74.83	3.684-35.60	8.25-11.67
Linear optical parameters						
Samples	α		E_i (eV)			
	200 nm< λ <400 nm	400 nm< λ <800 nm				
BaTiO_3 ^[110]	0.99	0.684	3.6			
(0.8)PVDF/(0.2)BaTiO ₃ ^[110]	0.999	0.997	2.9			
0.6PVDF/0.4BaTiO ₃ ^[110]	0.999	0.999	2.4			
1PVDF/5BaTiO ₃ ^[109]	0.9-0.5	0.2-0.5	3.85-3.3			

Note: P-PVDF, E_d -Direct band gap, E_i -Indirect optical band gap, n -Linear refractive index ($\lambda \approx 633$ nm), n_2 - nonlinear refractive index ($\text{cm}^2/\text{W} \times 10^{-12}$), T-transmittance ($\lambda \approx 633$ nm), $\chi^{(3)}$: third-order nonlinear optical susceptibility (10^{-8} esu), α -linear absorption coefficient, P_{th} -optical limiting threshold power (MW/cm^2), L_{eff} : the effective length (μm); $\Delta\phi_0$: the on-axis nonlinear phase shift at the focus; n_{∞} : the long wavelength refractive index, λ_0 : the average oscillator wavelength, S_0 : average oscillator strength (10^{14} m^{-2}), ϵ_{∞} : high-frequency dielectric constant, N/m^* : free carriers ratio ($\times 10^{17}$ m^{-3}), ω_p : the plasma frequency (10^{14} s^{-1}), n -Linear refractive index ($\lambda \approx 633$ nm), k -the extension coefficient (10^{-3} , 220 nm< λ <380 nm).

综上所述,无机非金属晶体材料中的 HNT、 $\text{Li}_4\text{Ti}_5\text{O}_{12}$ 等掺杂可以改善 PVDF 薄膜的非线性光学性能,尤其是薄膜的厚度、掺杂物的浓度与薄膜的光学带隙及透过率线性相关,直接决定着 NLO 效应及发光性能的优劣;金属盐中 MoS_2 有掺杂价值,辐照作用会导致薄膜带隙的降低;而多填料复合的方法,其主要作用为提高 PVDF 的 β 相含量,增强了复合材料薄膜的介电性能与极化能力。这些研究均可作为掺杂 PVDF 光学薄膜的制备提供参考。

3.4 PVDF 及其共聚物薄膜的量子化学计算

量子化学计算方法是研究晶体物理及化学性质的一种重要手段。对于铁电体材料,如果能从第一性原理出发,引入分子动力学等计算方法,必能更透彻地理解铁电体的自发极化、NLO 响应等微观机制。近年来,有不少学者利用第一性原理中的分子动力学方法,从光子带隙入手计算模拟 PVDF 的晶体结构并研究其 NLO 性质^[28,51,128-133]。文献 [131] 采用密度泛函理论 (density functional theory, DFT) 计算了包含 1250 个大原子系统的 Ag/PVDF 复合材料 (图 7(a)) 的光学性能参数, Ag 纳米颗粒的加入使薄膜的反射率 R 从 0 变为

40%,带隙值大于 1.6 eV 时其值与纯 Ag 纳米颗粒吸收值相近 (图 7(b)),由于入射光线不依赖于场方向,因此曲线重叠。Chun-gang Duan^[70] 基于第一性原理采用全势线性增强平面波 (FLAPW) 法计算 PVDF 均聚物的光子带隙及 n 、SHG 系数、LEO 系数 (线性电光) 等参数,取得了重要成果,其计算值与目前为止的实验值有较好的吻合 (图 7(d)、7(e))。文献 [69] 介绍了利用 DFT 理论计算 PVDF/(1~3) ZrO_2 薄膜的光子带隙、 n 、 $\chi^{(3)}$ 的变化情况, ZrO_2 掺杂量的倍增会使薄膜分子偶极矩增大,从而导致更高的非线性极化率,同时相应的介电常数及折射率均线性增大。

程和平等^[132] 利用含有色散修正的平面波超软赝势法 (基于 DFT) 计算得到了 PVDF(9 种晶型) 的光子带隙及 n 、 α 及 T 谱 (图 7(f)), 同时发现 PVDF 的光学性质的变化主要集中于深紫外等短波区域,而这一结论与众多实验结果均有较好吻合。文献 [130] 则通过 DFT 计算阐明了 PVDF 相变过程中晶体结构的二面角动态变化 (图 7(c)), 并解释了实验过程中观察到的 P(VDF-TrFE-CFE) 三元共聚物折射率及光程变化的原因,即聚合物材料体系中存在外场条件下的纳米

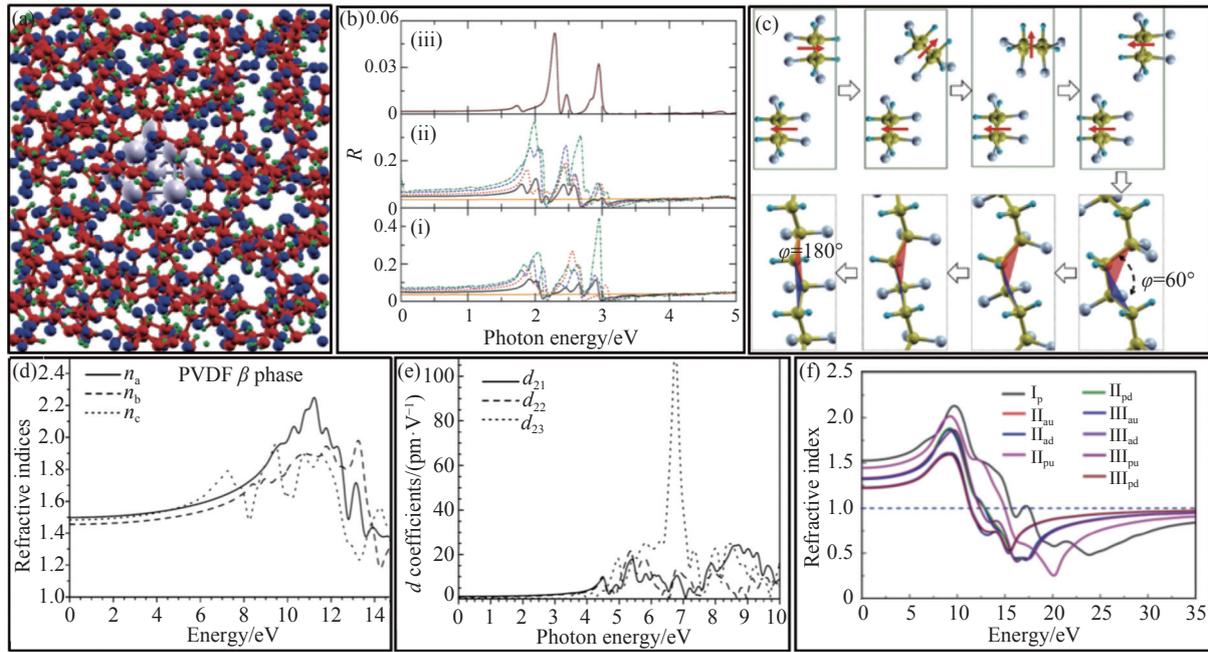


图 7 PVDF 及其共聚物薄膜的量子化学计算。(a) Ag/PVDF 复合材料 1250 大原子系统模拟元胞, 银原子灰色, 氟原子蓝色, 碳原子红色, 氢原子绿色^[131]; (b) Ag/PVDF 纳米复合材料的反射率 R : (i) 与 (ii) 表示入射光沿 z 轴与 x 轴, (iii) 表示真空中 Ag 纳米颗粒沿 z 轴和 x 轴入射^[131]; (c) 上图为非极性 α 相 \rightarrow 极性 γ 、 β 相的跃迁路径, 下图为 PVDF 的 α 相 TGTG 链转变为 β 相的全 T 链时, 分子链二面角变化, 红色箭头表示偶极矩的方向, 而黑色双箭头表示几何级数^[130]; (d) PVDF 的折射率与 (e) 色散计算值, a 、 b 、 c 分别代表光子沿 a 、 b 、 c 轴的极化方向^[70]; (f) PVDF (九相晶体结构) 折射率的理论计算值^[132]

Fig. 7 Quantum chemical calculation for PVDF and its copolymers films. (a) Snapshot of a Ag/PVDF1250 nanocomposites simulation cell. Silver atoms are shaded gray, fluorine blue, carbon red, and hydrogen green^[131]; (b) the normal reflectance (R) of Ag/PVDF nanocomposites materials: (i) and (ii) are incident light along the z - and x -axes; (iii) present the respective optical properties of Ag-nanoparticles in vacuum, with incident fields along the z - and x -axes^[131]; (c) the two-step transition pathway connecting the nonpolar α phase and the polar γ and β phases, the top panels show the conversion to the γ phase through a rotation of one chain, the bottom panel illustrates dihedral angle changes in the TGTG chain as it transforms to the T chain of the β phase. Red arrows indicate the directions of the dipole moments while black double arrows indicate the geometrical progression^[130]; (d) calculated dispersion curves for the d -coefficients and (e) the refractive indices of PVDF. Labels a , b , and c represent the photon polarization directions along the crystal axes a , b , and c , respectively;^[70] (f) theoretically calculated refractive index of PVDF^[132]

极化微区重新取向, 从而产生电致伸缩应变和 EO 效应。此外, 有研究人员^[133] 还利用第一性原理计算了共聚物 PVDF-HFP 及 PVDF-BTFE 的介电常数, 而南策文团队^[134] 则利用第一性原理并结合 PFM 及相场模拟等技术获得了 P(VDF-TrFE) 共聚物薄膜的拓扑态结构, 并成功测出其远红外区吸收谱特征。

基于第一性原理的光子带隙计算, 不仅可以较为精确地计算出折射率、消光系数等线性光学常数, 还能得到双折射、电光系数、二阶及三阶极化率等 NLO 参数, 而相关计算结果与实验结果都有较好的吻合, 这为 PVDF 及其共聚物薄膜的 NLO 相关研究提供了更为先进的手段, 并建立起

计算模拟-工艺制备-性能表征-结果验证的闭环研究范式。

4 结论与展望

本文定义了 PVDF 及其共聚物薄膜的概念, 并将其分为本征型与掺杂型两类, 着重从薄膜的制备、NLO 性能测试表征及量子化学计算几个方面进行了综述。可知: 电场、力场、温度场等外场施加, 使 PVDF 的极性相 (β 相) 及结晶度提升是实现其 NLO 效应的前提; 填料纳米化掺杂与共混则是目前为止成本最低、工艺最简单、有效提升其非线性光学性能的现实路径, 其中金属氧化物

及低维碳材料的掺杂效果最为突出;LB法为PVDF及其共聚物薄膜高极化率及纳米超薄化实现提供了重要途径;多种复合填料的选择、配比为PVDF及其共聚物薄膜的光学调控提供了有效方式;基于第一性原理的光子带隙计算为PVDF及其共聚物薄膜的晶体结构及NLO研究提供了理论支撑和闭环研究范式;高灵敏度、宽测量范围Z-扫描、马克条纹法结合椭圆偏振及图像处理技术可为PVDF及其共聚物薄膜的NLO测试提供主要测试手段。

总之,PVDF及其共聚物光学薄膜在强光频

电场或低频(直流)电场作用下,表现出EO、NLO等光学效应,但是目前来看,此类薄膜的NLO研究还不够充分。如制备nm尺度的薄膜工艺缺乏或不够成熟,掺杂类薄膜较多而本征型薄膜较少,测试手段以Z-扫描为主比较单一且精度有待进一步提高,NLO相关参数的理论计算还不够深入全面。如果能从这些薄弱点发力加强研究,定能加深对该类薄膜材料微观极化机制的理解认识,从而促成其在非线性光学、集成光学、光电子学及微纳光学等领域的广泛应用。

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